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## Crystal Structure of the Four-layer Orthorhombic Polytype of Potassium Hexacyanocobaltate(III)

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A four-layer orthorhombic polytype of  $K_3Co(CN)_6$  has been identified and the crystal structure determined by three-dimensional X-ray methods using 854 intensities obtained by counter methods with Mo  $K\alpha$  radiation at room temperature. The crystal is orthorhombic with space group  $Pnc2$  and cell constants of  $a=26.69$ ,  $b=10.353$  and  $c=8.34$  Å. There are eight formula units per unit cell. The interdependence of parameters associated with the Co and K atoms occupying similar positions in the various layers gave rise to such large correlation coefficients that it was impossible to refine the parameters of all the atoms in the asymmetric unit simultaneously. Two groups of non-interfering parameters were therefore refined separately by full-matrix least squares to a conventional  $R$  of 0.09. Bond lengths and bond angles of the different octahedra are given and a possible distortion is discussed in terms thereof.

### Introduction

A crystal of Cr-doped  $K_3Co(CN)_6$  has been used in a number of nuclear magnetic resonance (n.m.r.) experiments in which, among other things, the second moment (Lourens & Reynhardt, 1971a) and the quadrupole interaction (Lourens & Reynhardt, 1971b) of the  $^{59}Co$  nuclei were measured. With a view to the interpretation of these results in terms of the atomic environments of the  $^{59}Co$  nuclei, the determination of this structure was undertaken, since a review of the available crystallographic literature on  $K_3Co(CN)_6$  shows complete confusion.

This confusion was partially resolved by Kohn & Townes (1961) who first demonstrated the existence of the  $1M$ ,  $20r$ ,  $3M$  and  $7M$  polytypes. The  $40r$  polytype should now be added to these. The early investigators obviously dealt with several of these and at-

tempted to assign a single space group to explain all their observations.

In this way Barkhatov & Zhdanov (1942) gave cell dimensions very similar to those for the  $40r$  polytype, but by analogy with the results of Gottfried & Nagelschmidt (1930) for  $K_3Fe(CN)_6$  they assigned the space group  $P2_1/c$  and called it pseudo-orthorhombic. Our impression is that these workers actually encountered the  $40r$  type and not only the  $1M$  type as suggested by Kohn & Townes (1961). Okaya & Pepinsky (1956), who first described the  $20r$  polytype correctly, also attempted to explain the earlier results (obtained for other polytypes) in terms of their observations.

Barkhatov (1942) proposed the first structure based on the unit cell of the  $1M$  polytype. In their neutron-diffraction study Curry & Runciman (1959) accepted the special positions of the cobalt and potassium atoms suggested by Barkhatov (1942) and determined the struc-

ture in (001) projection. Values for the  $z$  coordinates of the carbon and nitrogen atoms were calculated geometrically by assuming that the  $\text{Co}(\text{CN})_6^{3-}$  groups form perfect octahedra. It was, however, impossible to calculate the  $z$  coordinate of the potassium atom in a general position.

### Sample preparation

The crystal used in this investigation was grown by Rennie & Nielsen (1959) from an aqueous solution of  $\text{K}_3\text{Co}(\text{CN})_6$  containing 0.6% potassium chromicyanide and 0.1% potassium sulphate. They used a dropping temperature method. The addition of the potassium sulphate results in the growth of crystals with large  $b$  faces.

An atomic absorption analysis indicated that the crystal contains 0.079% chromium (per weight).

### Experimental

A spherical crystal, 0.4 mm in diameter, was mounted on a goniometer with the  $c$  axis as rotation axis. Preliminary oscillation and Weissenberg photographs suggested a one-layer monoclinic structure with space group  $P2_1/c$  (No. 14 in *International Tables for X-ray Crystallography*, 1965) as described by Curry & Runciman (1959).

The crystal was mounted on a Hilger & Watts single-crystal diffractometer. Intensity data for the sphere with  $\sin \theta/\lambda \leq 0.482$  were collected using  $\text{Mo } K\alpha$  radiation. An  $\omega-2\theta$  scan mode was used, counting for a total of 200 sec per reflexion. This was done in 40 steps of  $0.02^\circ$  and counting for 5 sec per step. A total of 1786 reflexions were measured. 701 of these were not significantly above the background. The determination of the cell dimensions by a method of least squares yielded the following results:  $a=6.99$ ,  $b=10.353$ ,  $c=8.34 \text{ \AA}$  and  $\beta=107.33^\circ$ . The density calculated for two formula units per unit cell is  $1.84 \text{ g.cm}^{-3}$ . This value compares well with the measured density of  $1.81 \text{ g.cm}^{-3}$ . Corrections for absorption, Lorentz and polarization effects were applied using a standard computer program. All Fourier syntheses were calculated by using the centrosymmetric Fourier program of P. Gantzel & H. Hope.

A three-dimensional Patterson synthesis was carried out and found to be fully consistent with the one-layer structure proposed by Curry & Runciman (1959). A three-dimensional Fourier synthesis phased on the heavy-atom positions obtained in this way, however, did not fit this structure at all. Despite all efforts to the contrary, a coordination octahedron like that around the Co atom at (000) persistently occurred around the K atom at  $(00\frac{1}{2})$ . This suggested either a disordered structure or a polytype as described by Kohn & Townes (1961), but with one half of the layers shifted through half a cell translation along  $c$ .

An ( $h1l$ ) precession photograph, similar to those

taken by Kohn & Townes (1961) to identify the different polytypes, was taken. This photograph revealed a four-layer orthorhombic structure not previously reported. The previously measured intensities were examined carefully to confirm that the crystal system was in fact orthorhombic.

The new cell dimensions are:  $a=26.69$ ,  $b=10.353$ ,  $c=8.34 \text{ \AA}$ .

The length of the  $a$  axis is twice the length of the corresponding axis of the  $20r$  polytype described by Okaya & Pepinski (1956). The  $h$  indices of the measured intensities were transformed to fit the new cell. This reduced the number of observed intensities to 699.

The same crystal was mounted on the diffractometer to measure some of the extra reflexions. Since these reflexions were almost invisible on Weissenberg photographs, it was decided to double the total counting time per intensity. 299 intensities (144 of these were unobserved) for the sphere with  $\sin \theta/\lambda \leq 0.245$  were collected and put on the same scale as the other set. The same corrections as before were made. Altogether 854 observed intensities were collected.

### Determination of the structure

The reflexions which define the  $40r$  rather than the  $1M$  lattice occur only in limited and special regions and so give rise to the occurrence of a large number of accidental absences. Because of this no decision could be reached as to whether the  $hk0$  reflexions were systematically or accidentally absent for  $h$  odd. It was therefore necessary to consider both of the space groups  $Pnca$  (No. 60) and  $Pncm$  (No. 53) or the non-centrosymmetric  $Pnc2$  (No. 30 in *International Tables for X-ray Crystallography*, 1965).  $Pnca$ , however, leads to exactly the same problem encountered with  $P2_1/c$ . It cannot fully account for the observed electron densities since there are no symmetry elements which can correctly relate the mutually shifted subcells.

Since  $Pnca$  is the space group of the  $20r$  variety (Okaya & Pepinsky, 1956) an attempt was nevertheless made to use this space group in a least-squares refinement, but the calculated structure factors, particularly for 111 which remained very high in comparison with the observed value, showed that the space group assignment was wrong. The minimum  $R$  was 0.14. The space group  $Pncm$  fitted neither the structure which was known in outline from our attempts with  $P2_1/c$  nor the orientation of the hexacyanocobaltate octahedra as derived from n.m.r. experiments (Lourens & Reynhardt, 1971a). According to these there are four magnetically inequivalent cobalt sites in the unit cell. This implies four differently oriented octahedra. These four types are present in two main groups, those in one group being mirror images across the  $ac$  plane of those in the other group. Each of these two main groups consists of two types of octahedra, the one differing only slightly from the other. The view was taken that the two groups of octahedra described by Kohn & Townes

(1961), represent the two main groups. The only remaining space group and the only one which could be fitted successfully to the trial structure was  $Pnc2$ .

The program *ORFLS* of Busing, Martin & Levy (1962) was used to do a least squares refinement of atomic parameters of the trial structure on an IBM 360/65 computer. It was impossible to refine the parameters of all 34 atoms in the asymmetric unit simultaneously. This problem relates to the fact that the four subcells in the four-layer structure are very much alike. In the trial structure both Co and K atoms occurred in special positions  $a(0,0,z)$  and  $b(\frac{1}{2},0,z)$  of the space group with  $z(\text{Co})=0$  and  $z(\text{K})=\frac{1}{2}$ . Furthermore Co and K atoms also occurred in quasi-special positions of the type  $\frac{1}{4}, 0, \frac{1}{4}$  and  $\frac{1}{4}, 0, \frac{3}{4}$  respectively. Since Co and K atoms have similar scattering powers these atoms appear to be related by non-space group symmetry and, (Stout & Jensen, 1968), large correlation coefficients occurred among their parameters when least squares refinement was attempted. This caused the results of successive cycles to oscillate severely and eventually to diverge. It was therefore necessary to refine the parameters of these atoms separately, one atom at a time. The rest of the parameters could be refined simultaneously.

Table 1. Fractional coordinates ( $\times 10^4$ ) and isotropic temperature factors of the atoms in the asymmetric unit

Standard deviations are given in parentheses.

	$x$	$y$	$z$	$B$ ( $\text{\AA}^2$ )
Co(1)	0	0	0	1.31 (35)
Co(2)	$\frac{1}{2}$ *	0	40 (20)	1.71 (39)
Co(3)	7506 (3)	8 (7)	2496 (13)	1.68 (38)
K(1)	0	0	4990 (14)	1.93 (37)
K(2)	$\frac{1}{2}$ *	0	4965 (12)	1.44 (40)
K(3)	7505 (12)	9 (10)	7515 (20)	1.47 (35)
K(4)	1272 (15)	2913 (20)	53 (26)	1.69 (40)
K(5)	1252 (11)	7264 (18)	2438 (33)	1.60 (12)
K(6)	3715 (14)	2560 (23)	2463 (26)	2.07 (28)
K(7)	3714 (19)	7357 (16)	73 (36)	1.65 (33)
C(1)	470 (23)	502 (51)	1560 (61)	1.49 (44)
C(2)	-288 (24)	1678 (64)	13 (70)	1.28 (29)
C(3)	476 (21)	515 (52)	-1554 (63)	1.52 (47)
C(4)	5467 (12)	510 (64)	-1550 (74)	1.37 (38)
C(5)	4708 (20)	1677 (52)	-1 (65)	1.54 (42)
C(6)	5481 (31)	524 (70)	1572 (71)	1.95 (50)
C(7)	7981 (54)	486 (61)	4104 (81)	1.58 (38)
C(8)	7021 (21)	-492 (41)	911 (62)	1.33 (32)
C(9)	7033 (12)	-522 (42)	4030 (62)	1.05 (31)
C(10)	7971 (22)	512 (52)	967 (74)	1.85 (30)
C(11)	7218 (21)	1671 (63)	2560 (71)	1.46 (21)
C(12)	7779 (23)	-1663 (54)	2446 (73)	1.28 (15)
N(1)	772 (25)	808 (61)	2512 (73)	1.13 (41)
N(2)	-478 (21)	2706 (51)	21 (71)	2.22 (39)
N(3)	767 (20)	837 (53)	-2505 (84)	2.21 (26)
N(4)	5748 (13)	822 (46)	-2496 (60)	1.05 (33)
N(5)	4524 (13)	2697 (41)	-9 (52)	2.13 (35)
N(6)	5775 (21)	858 (72)	2530 (81)	2.66 (49)
N(7)	8272 (22)	783 (45)	5060 (93)	1.43 (51)
N(8)	6725 (13)	-777 (54)	-28 (64)	2.65 (40)
N(9)	6756 (21)	-831 (51)	5003 (80)	1.31 (38)
N(10)	8255 (12)	823 (40)	-80 (81)	1.28 (20)
N(11)	7046 (31)	2677 (61)	2597 (91)	3.41 (51)
N(12)	7959 (21)	-2671 (53)	2417 (72)	1.04 (43)

\* Fractional coordinates are not multiplied by  $10^4$ .

Apart from fractional atomic coordinates, isotropic individual temperature factors were refined until the  $R$  value ( $\sum|\Delta F|/\sum F_0$ ) reached 0.090. No attempt was made to introduce anisotropic thermal parameters in the refinement. The fractional atomic parameters, together with the individual isotropic temperature factors, are given in Table 1, while the calculated and observed structure factors, on an absolute scale, are shown in Table 2.

### Description and discussion of the structure

The structure is shown in projection along two crystallographic axes in Fig. 1.

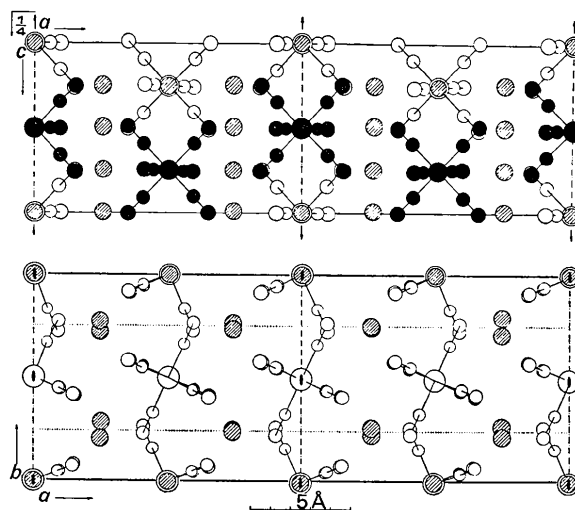


Fig. 1. (001) and (010) projections of the structure. The striped and big circles represent potassium and cobalt atoms, respectively. The bonding sequence is always Co-C-N. Atoms represented by solid circles are associated with molecules lying half a cell translation above the paper.

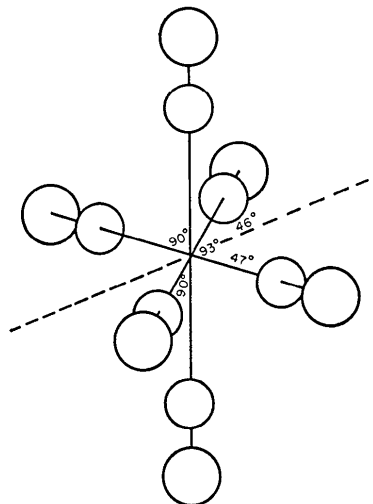


Fig. 2. Distortion of the  $K_3Co(CN)_6$  octahedra as averaged over the three crystallographically distinct ions.



These distances agree well with other values which have been quoted in the literature. The average Co-C bond length calculated by Curry & Runciman (1959) is 1.89 Å. Pauling, Springall & Palmer, (1939) found a C-N distance of  $1.16 \pm 0.02$  Å in methyl cyanide and Monfort (1942) found a similar value in potassium sodium platinumocyanide.

Table 3. *Observed molecular bond lengths in Å*

Standard deviations are given in parentheses.

		Average
Co(1)-C(1)	1.881 (11)	
Co(1)-C(2)	1.900 (11)	1.89 (1)
Co(1)-C(3)	1.892 (11)	
Co(2)-C(4)	1.872 (11)	
Co(2)-C(5)	1.903 (11)	1.90 (1)
Co(2)-C(6)	1.913 (12)	
Co(3)-C(7)	1.921 (12)	
Co(3)-C(8)	1.911 (11)	
Co(3)-C(9)	1.864 (11)	1.89 (1)
Co(3)-C(10)	1.870 (11)	
Co(3)-C(11)	1.887 (11)	
Co(3)-C(12)	1.876 (11)	
C(1)-N(1)	1.175 (12)	
C(2)-N(2)	1.179 (12)	1.17 (1)
C(3)-N(3)	1.158 (12)	
C(4)-N(4)	1.136 (12)	
C(5)-N(5)	1.165 (12)	1.16 (1)
C(6)-N(6)	1.172 (13)	
C(7)-N(7)	1.155 (13)	
C(8)-N(8)	1.150 (12)	
C(9)-N(9)	1.144 (12)	
C(10)-N(10)	1.200 (12)	1.16 (1)
C(11)-N(11)	1.139 (13)	
C(12)-N(12)	1.148 (12)	

The Co-C-N bonds are almost collinear, the maximum deviation being  $2^\circ$ . As can be seen from Table 4, there is for each of the octahedra, one C-Co-C bond angle which differs appreciably ( $\sim 3^\circ$ ) from a right angle. It can be described as a digonal distortion with the crystallographic  $c$  axis as the digonal axis. The

average distortion for the three distinct octahedra is indicated in Fig. 2. Although it is known from n.m.r. results (Sugawara, 1959) that at the Co nuclei there are asymmetric electric field gradients which have their origin in the asymmetry of the  $Co(CN)_6$  groups, the fact that the refinement of this structure was done in two different cycles could make this result somewhat contentious.

The stacking mechanism can only formally be described in terms of the rotations suggested by Kohn & Townes (1961). The most useful description to our minds is in terms of four monoclinic subcells with two adjacent ones, as shown in Fig. 3, shifted through  $c/2$  along  $c$ . This is fully consistent with a step-growth mechanism (Frank, 1952) which leads to the occurrence of various polytypes depending on the height of an exposed screw-dislocation ledge and it seems likely that all the other polytypes of  $K_3Co(CN)_6$  have structures of this same type.

The authors are grateful to Mr K. Kröger of the Spectrochemistry Division of the C.S.I.R. for the atomic absorption analysis of the crystal. Thanks are also due to Professor J.A.J. Lourens, University of South Africa for his stimulating interest in this work and for suggesting the problem.

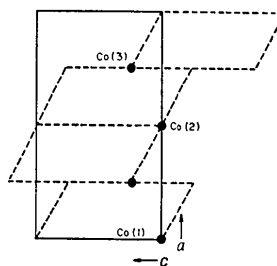


Fig. 3. Diagram to illustrate the relationship between the  $1M$  and  $40r$  unit cells.

Table 4. *Observed bond angles in degrees*

Standard deviations are given in parentheses.

C(1)-Co(1)-C(2)	91.0 (7)	C(4)-Co(2)-C(5)	90.8 (7)
C(1)-Co(1)-C(3)	87.0 (7)	C(4)-Co(2)-C(6)	86.9 (7)
C(2)-Co(1)-C(3)	90.8 (7)	C(5)-Co(2)-C(6)	90.9 (7)
C(1)-Co(1)-C(1')*	92.4 (7)	C(4)-Co(2)-C(4')	92.6 (7)
C(2)-Co(1)-C(1')	88.8 (7)	C(5)-Co(2)-C(4')	89.1 (7)
C(2)-Co(1)-C(3')	89.4 (7)	C(5)-Co(2)-C(6')	89.1 (7)
C(3)-Co(1)-C(3')	93.5 (6)	C(6)-Co(2)-C(6')	93.5 (6)
C(7)-Co(3)-C(11)	90.4 (7)	Co(1)-C(1)-N(1)	178.6 (1.2)
C(9)-Co(3)-C(7)	93.4 (7)	Co(1)-C(2)-N(2)	178.5 (1.2)
C(9)-Co(3)-C(11)	89.0 (7)	Co(1)-C(3)-N(3)	179.7 (1.2)
C(7)-Co(3)-C(10)	87.3 (7)	Co(2)-C(4)-N(4)	179.6 (1.2)
C(7)-Co(3)-C(12)	89.5 (7)	Co(2)-C(5)-N(5)	179.1 (1.2)
C(8)-Co(3)-C(9)	87.1 (7)	Co(2)-C(6)-N(6)	178.5 (1.2)
C(8)-Co(3)-C(10)	92.9 (7)	Co(3)-C(7)-N(7)	179.5 (1.2)
C(8)-Co(3)-C(11)	89.8 (7)	Co(3)-C(8)-N(8)	178.7 (1.2)
C(8)-Co(3)-C(12)	90.2 (7)	Co(3)-C(9)-N(9)	178.0 (1.2)
C(9)-Co(3)-C(12)	90.9 (7)	Co(3)-C(10)-N(10)	178.0 (1.2)
C(10)-Co(3)-C(11)	91.5 (7)	Co(3)-C(11)-N(11)	179.7 (1.2)
C(10)-Co(3)-C(12)	88.9 (7)	Co(3)-C(12)-N(12)	178.7 (1.2)

\* Atoms (1) and (1') have fractional coordinates  $(x, y, z)$  and  $(\bar{x}, \bar{y}, z)$  respectively.

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## Die Kristallstrukturen von Hexamminchrom(III)-Hexafluoromanganat(III) und Hexamminchrom(III)-Hexafluoroferrat(III)

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The crystal structures of  $[\text{Cr}(\text{NH}_3)_6]\text{MnF}_6$  and of  $[\text{Cr}(\text{NH}_3)_6]\text{FeF}_6$  have been determined from three-dimensional X-ray data collected by counter methods. The structures have been refined by full-matrix least-squares techniques to final conventional  $R$ -values of 3.0% for 294 independent reflexions for  $[\text{Cr}(\text{NH}_3)_6]\text{MnF}_6$  and of 3.4% for 648 independent reflexions for  $[\text{Cr}(\text{NH}_3)_6]\text{FeF}_6$ . The compounds crystallize in the cubic space group  $Pa\bar{3}$  ( $T_h^2$ ), with four formula units in a cell ( $a = 10.059(3)$  Å for  $[\text{Cr}(\text{NH}_3)_6]\text{MnF}_6$  and  $10.079(3)$  Å for  $[\text{Cr}(\text{NH}_3)_6]\text{FeF}_6$ ). A dynamical Jahn-Teller effect for the  $\text{MnF}_6^{3-}$  ion is discussed. Evidence for hydrogen bonding in complex hexafluorides containing the  $[\text{Cr}(\text{NH}_3)_6]^{3+}$  ion is given.

### Einleitung

Röntgenographische und infrarot-spektroskopische Untersuchungen an den Verbindungen  $\text{K}_3\text{MnF}_6$ ,  $\text{K}_2\text{NaMnF}_6$  und  $\text{Cs}_2\text{KMnF}_6$  ergaben  $D_{4h}$ -Symmetrie der  $\text{MnF}_6^{3-}$ -Anionen. Sechs Fluoratome umgeben  $\text{Mn}^{3+}$  in Form eines gestreckten Oktaeders (Peacock, 1957; Knox, 1963; Wieghardt & Siebert, 1971; Schneider & Hoppe, 1970). Diese statische Verzerrung des Komplexoktaeders wird als eine Folge des Jahn-Teller Theorems (Jahn & Teller, 1937) gedeutet.

Die kürzlich dargestellten Salze  $[\text{M}(\text{NH}_3)_6]\text{MnF}_6$  ( $\text{M} = \text{Cr, Co, Rh}$ ) kristallisieren in der kubischen Raumgruppe  $T_h^2-Pa\bar{3}$ . Die Lagesymmetrie der  $\text{MnF}_6^{3-}$ -Anionen ist  $C_{3i}$  ( $S_6$ ). Das bedeutet aber, dass die Symmetrie der  $\text{MnF}_6^{3-}$ -Ionen in diesen Salzen nicht  $D_{4h}$  sein kann (Wieghardt & Siebert, 1971a). Da sich auch die Infrarot-Spektren wesentlich von denen der  $\text{K}_3\text{MnF}_6$ - und  $\text{K}_2\text{NaMnF}_6$ -Salze unterscheiden (s.unten), wurde die Kristallstruktur von  $[\text{Cr}(\text{NH}_3)_6]\text{MnF}_6$  bestimmt.

Eine zweite Kristallstrukturanalyse wurde von dem isotypen Salz  $[\text{Cr}(\text{NH}_3)_6]\text{FeF}_6$  durchgeführt, da das  $\text{Fe}^{3+}$ -Ion keinen Jahn-Teller Effekt zeigt. Es eignet sich daher gut zu einem Vergleich mit  $[\text{Cr}(\text{NH}_3)_6]\text{MnF}_6$ .

### Experimentelles

$[\text{Cr}(\text{NH}_3)_6]\text{MnF}_6$  und  $[\text{Cr}(\text{NH}_3)_6]\text{FeF}_6$  wurden nach einer von Wieghardt & Siebert (1971a, b) beschriebenen Methode dargestellt. Die Salze kristallisieren in kleinen, gut ausgebildeten Würfeln, die optisch isotrop sind. Drehkristall- und Weissenberg-Aufnahmen ergaben kubische Symmetrie. Aufgrund der systematischen Auslöschungen  $0kl$  für  $k = 2n + 1$ ,  $h0l$  für  $l = 2n + 1$ ,  $hk0$  für  $h = 2n + 1$  und der gleichen Intensität aller Reflexe mit cyclisch vertauschbaren Indizes kommt nur die Raumgruppe  $T_h^2-Pa\bar{3}$  in Frage.

Die Gitterkonstanten wurden aus diffraktometrisch bestimmten Winkeln  $\theta$  von jeweils 2 Reflexen  $h00$ ,  $0k0$ ,  $00l$  berechnet. In Tabelle 1 sind die Kristalldaten zu-